Lattice model for parallel and orthogonal β sheets using hydrogenlike bonding

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(Received 12 June 2007; published 6 November 2007)

We present results for a lattice model of polymers where the type of β sheet formation can be controlled by different types of hydrogen bonds depending on the relative orientation of close segments of the polymer. Tuning these different interaction strengths leads to low-temperature structures with different types of orientational order. We perform simulations of this model and so present the phase diagram, ascertaining the nature of the phases and the order of the transitions between these phases.

DOI: 10.1103/PhysRevE.76.051904

PACS number(s): 87.15.By, 05.70.Fh, 61.41.+e, 82.35.Lr

I. INTRODUCTION

The transition of a flexible macromolecular chain from a random-coil conformation to a globular compact form, called coil-globule transition, has been a subject of extensive theoretical and experimental studies [1]. Generally, polymers in a good solvent are modeled by random walks with short-range repulsion (excluded volume). Polymers undergoing a coilglobule transition are then modeled by adding an additional short-range attraction. This short-range attraction is both due to an affinity between monomers and solvent molecules, affecting the solvability of a polymer, and also due to intramolecular interactions between different monomers, for example due to van der Waals forces. The canonical lattice model [2,3] for this transition is given by interacting selfavoiding walks, in which self-avoiding random walks on a lattice are weighted according to the number of nearestneighbor contacts (nonconsecutively visited nearest-neighbor lattice sites).

In biological systems, e.g., proteins, the question of focus is usually the ground state of a polymer with a specified composition rather than the thermodynamic phase transitions of polymers in solution. Here, the most relevant contribution to monomer-monomer interactions is due to hydrogen bonds. These hydrogen bonds can only form if neighboring segments are aligned in certain ways, resulting in an interaction that is strongly dependent on the relative orientation of segments. This type of interaction plays a leading role in the formation of secondary protein structures such as α helices and β sheets [4]. Recent work [5–8] has focused on the variety of different protein structures that can be designed when using various types of hydrogenlike bonding in conjunction with other types of interactions for finite length polymers.

In this paper, we consider the thermodynamic phase structure of a lattice polymer model with competing types of hydrogenlike bonding, rather than the complicated ground states of short-length configurations.

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interacting via hydrogen bonds, in which hydrogen bonds were mimicked by an interaction between two nearestneighbor lattice sites which belong to two straight segments of the polymer. This was treated in the context of Hamiltonian walks in a mean-field approach, and they predicted a first-order transition between an anisotropic ordered phase and a molten phase. Later, Foster and Seno introduced this type of interaction to a model of self-avoiding walks [10]. They analyzed it using transfer-matrix techniques in two dimensions, where a first-order transition between a folded polymer crystal and a swollen coil was found. Subsequently, a variant of this model was introduced by Buzano and Pretti [11], where the interaction is defined between parallel nearest-neighbor bonds, independent of the straightness required in [9], arguing that these should better take into account the contribution of fluctuating bonds, which may be formed even in relatively disordered configurations. The authors studied this interacting-bond model and the one introduced by Foster and Seno on the square and simple cubic lattice using the Bethe approximation. They found a firstorder transition in the Foster-Seno model in two and three dimensions, confirming and extending results in [10]. In contrast to this, they found two transitions in the interactingbond model, a second-order θ transition from a swollen coil to a collapsed molten globule and then a first-order transition to a folded polymer crystal. In a later paper [12], they introduced a competing isotropic interaction and studied its effect in three dimensions using the Bethe approximation. They found a phase diagram with three different phases (swollen coil, collapsed molten globule, folded polymer crystal), similar to that of collapsing semistiff polymers [13]. In this work we generalize the Foster-Seno model to dis-

In [9], Bascle *et al.* introduced a lattice model of polymers

In this work we generalize the Foster-Seno model to distinguish between nearest-neighbor contacts of parallel and orthogonal straight segments (see Fig. 1) and assign interactions of different strengths to these two types of contacts, investigating it with Monte Carlo simulations using the Flat-PERM algorithm [14]. We begin by simulating the Foster-Seno model and confirm the theoretical picture presented above [10,12]. We then consider our extended model (in three dimensions). We find evidence for two differently structured folded phases, depending on whether the parallel or orthogonal interactions dominate. The transition between the swollen coil and each of the two collapsed ordered crys-

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FIG. 1. (Color online) The two types of nearest-neighbor contacts between two straight segments of the polymer: orthogonal segments (a) with interaction $-\varepsilon_o$, and parallel segments (b) with interaction $-\varepsilon_p$. In two dimensions, only parallel interactions are possible.

tals is first-order. We investigate the structure of these two low-temperature phases. For strong parallel interactions long segments of the polymer align, whereas for strong orthogonal interactions the polymer forms alternating orthogonally layered β sheets.

II. MODEL AND SIMULATIONS

A polymer is modeled as an *n*-step self-avoiding walk on the simple cubic lattice with interactions $-\varepsilon_p$ and $-\varepsilon_o$ for nearest-neighbor contacts between parallel and orthogonal *straight* segments of the walk, as shown in Fig. 1. Here, a segment is defined as a site along with the two adjoining bonds visited by the walk, and we say that a segment is straight if these two bonds are aligned. The restriction of this model to $\varepsilon_o = \varepsilon_p$ is the simple generalization of the Foster-Seno model, which was originally defined on a square lattice, to three dimensions.

The total energy for a polymer configuration φ_n with n+1 monomers (occupied lattice sites) is given by

$$E_n(\varphi_n) = -m_p(\varphi_n)\varepsilon_p - m_o(\varphi_n)\varepsilon_o \tag{1}$$

depending on the number of nonconsecutive parallel and orthogonal straight nearest-neighbor segments m_p and m_o , respectively, along the polymer. For convenience, we define

$$\beta_p = \beta \varepsilon_p \quad \text{and} \quad \beta_o = \beta \varepsilon_o,$$
 (2)

where $\beta = 1/k_B T$ for temperature *T* and Boltzmann constant k_B . The partition function is given by

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$$Z_n(\beta_p, \beta_o) = \sum_{m_p, m_o} C_{n, m_p, m_o} e^{\beta_p m_p + \beta_o m_o}$$
(3)

with C_{n,m_p,m_q} being the density of states. We have simulated this model using the FlatPERM algorithm [14]. The power of this algorithm is the ability to sample the density of states uniformly with respect to a chosen parametrization, so that the whole parameter range is accessible from one simulation. In practice, we have also performed multiple independent simulations to further reduce errors. The natural parameters for this problem are m_p and m_o , and the algorithm directly estimates the density of states C_{n,m_p,m_o} for all $n \le n_{max}$ for some fixed n_{max} and all possible values of m_p and m_o . Canonical averages are performed with respect to this density of states. As we need to store the full density of states, we only perform simulations up to a maximal length of n_{max} = 128, due to a memory requirement growing as n^3 . To reduce the error, we have taken averages of ten independent runs each. Each run has taken approximately 3 months on a 2.8-GHz PC to complete.

Fixing one of the parameters β_p and β_o reduces the size the histogram, and enables us to perform simulations of larger systems, as the memory requirement now grows as n^2 . Fixing β_o , say, the algorithm directly estimates a partially summed density of states

$$\hat{C}_{n,m_p}(\beta_o) = \sum_{m_o} C_{n,m_p,m_o} e^{\beta_o m_o}.$$
(4)

In this way, we can simulate lengths up to $n_{max}=1024$ at fixed β_o . In a similar fashion, we also consider the diagonal $\beta_p = \beta_o = \beta$, which is equivalent to considering the partially summed density of states

$$\widetilde{C}_{n,m} = \sum_{m_o + m_p = m} C_{n,m_p,m_o}.$$
(5)

To reduce the error for our runs up to n=1024, we have taken averages of ten independent runs each. Each run has taken approximately 2 months on a 2.8-GHz PC to complete.

III. RESULTS

Before presenting the findings for our model, we briefly discuss the results of simulations of the Foster-Seno model in two dimensions. We find a first-order transition between a swollen coil and ordered collapsed phase in agreement with Foster and Seno [10]. Figure 2 shows the internal density distribution at $\beta = \beta_c = 1.04$, where the specific heat is maximal. This distribution is clearly bimodal, and finite-size scaling supports the conclusion that the transition is first-order. Our estimate of $\beta_c = 1.04$ is close to the value 1.00(2) obtained by Foster and Seno [10] from transfer matrix calculations. The low temperature phase is an ordered β -sheet type phase.

For the three-dimensional model, we have explored the full two-variable phase space (β_p, β_o) by using a two-



FIG. 2. (Color online) Internal energy density distributions for the two-dimensional Foster-Seno model at the value of β for which the fluctuations are maximal, length 1024.

parameter FlatPERM simulation of the model for lengths up to 128. We performed ten independent simulations to ensure convergence and understand the size of the statistical error in our results. As in previous work [15,16], we found the use of the largest eigenvalue of the matrix of second derivatives of the free energy with respect to the parameters β_p and β_o most advantageous to show the fluctuations in a unified manner. Figure 3 displays a density plot of the size of fluctuations for $0 \le \beta_p$, $\beta_o \le 2$. It suggests the presence of three thermodynamic phases separated by three phase transition lines meeting at a single point. For small values of β_p and β_o , we expect the model to be in the excluded volume universality class of swollen polymers, since at $\beta_p = \beta_o = 0$ the model reduces to the simple self-avoiding walk. The question arises as to the nature of the phases for large β_p with β_o fixed and for large β_o with β_p small, and the type of transitions between each of the phases.

We find evidence for a strong collapse phase transition when increasing β_p for fixed $\beta_o \leq 1.38$. Corrections to scaling at lengths $n \leq 128$ make it difficult to identify the nature of the transition. The location of the transition seems independent of the value of β_o and is located at $\beta_p \approx 1.25$ for length n=128; this is taken from the location of the peak of the fluctuations. Since our data indicate that this transition occurs for $\beta_o \leq 1.38$ at $\beta_p \approx 1.25$, it follows that the diagonal



FIG. 4. (Color online) A plot of the internal energy distribution in m_o at $\beta_p = 1.0$ for length n = 128 at values of β_o for which the fluctuations in m_o are maximal.

line $\beta_o = \beta_p$ crosses this transition line. Configurations in the collapsed phase are rich in parallel contacts; we shall discuss further details of the collapsed phase below.

The situation changes significantly for $\beta_o \ge 1.38$. When we start from the swollen phase at fixed $\beta_o > 1.38$ and increase β_p we see evidence for a strong phase transition to a different collapsed phase, in which orthogonal contacts are expected to play an important role. Further increase of β_p leads to another strong transition to the parallel-contact rich phase. We investigate the transition between the swollen coil and the orthogonal-contact rich phase by considering the line $\beta_p=1.0$. Figure 4 shows a bimodal internal energy distribution at the maximum of the fluctuations in m_o for length n=128, indicating the presence of a first-order transition.

Combining the evidence above, we conjecture the phase diagram shown in Fig. 5, having three phases and three transition lines that meet at a triple point located at $(\beta_p^t, \beta_o^t) \approx (1.25, 1.38)$ for length n=128. By considering the location of this point for different lengths n, we conclude that its estimate is affected by strong finite-size corrections to scaling.

To further elucidate the nature of the phase transitions and the structure of the low-temperature phases, we perform simulations for larger system sizes for the two lines $\beta_o=0$ and $\beta_o=\beta_p$, using one-parameter FlatPERM simulations for

orthogonal

3-sheets

 β_p

parallel

β-lines



FIG. 5. (Color online) This figure represents our conjectured schematic phase diagram. The phase boundaries are marked by solid (black) lines. The dashed diagonal (blue) and horizontal (red) lines denote the lines along which we have performed oneparameter simulations.

triple point

swollen coi

 β_o

FIG. 3. (Color online) This is a density plot of the logarithm of the largest eigenvalue of the matrix of second derivatives of the free energy with respect to β_p and β_o at n=128. The lighter the shade, the larger the value.



FIG. 6. (Color online) Internal energy density distributions of m_p at $\beta_p=0$ and $\beta_p=0.996$ for length 1024.

lengths up to n=1024, averaged over ten independent simulations each. We begin by considering $\beta_o = 0$. The peak of the specific heat occurs at $\beta_p = 0.996$ for n = 1024, which we note is shifted away from the value at length n=128 and reflects the presence of strong corrections to scaling. The distribution of m_p at this point is shown in Fig. 6; we observe a clear bimodal distribution with well-separated peaks and which ranges over fourteen orders of magnitude, convincingly supporting the conclusion of a first-order phase transition. Similarly, along the line $\beta = \beta_0 = \beta_p$ we find a single peak of the specific heat, located at $\beta_p = 0.998$ for n = 1024. The distribution of $m = m_o + m_p$ at this point displays the same characteristics as the transition on the line $\beta_0 = 0$ described above. Our investigations of the transition between the two collapsed phases were not conclusive, as it is difficult to do simulations at very low temperatures. While we expect there to be a first-order phase transition between the two collapsed phase we were unable to verify this.

To delineate the nature of the two collapsed phases, we have randomly sampled typical configurations for each: two of these are shown in Fig. 7. In each case we have used $\beta_p > 0$, where parallel contacts are attractive. For large β_p , we have a parallel contact rich phase, and typical configurations have lines of monomers arranged in parallel. In Fig. 7, there is a typical configuration for ($\beta_p = 1.8$, $\beta_o = 1.0$), which demonstrates these parallel β lines. For large β_o , orthogonal contacts play an important role. A typical configuration for $(\beta_p = 1.3, \beta_o = 1.9)$ consists of parallel lines arranged in β sheets, which are layered orthogonally. The entropy of the phase consisting out of orthogonal β sheets is lower than the entropy of the phase consisting out of collection of parallel lines, which explains why the collapse-collapse transition line is shifted away from the diagonal. Clearly the formation of β sheets is dependent on β_p being positive (attractive parallel) interactions.



FIG. 7. (Color online) Typical configurations for the two different collapsed phases, sampled at (a) ($\beta_p = 1.3$, $\beta_o = 1.9$) and at (b) ($\beta_p = 1.8$, $\beta_o = 1.0$).

In conclusion, we have demonstrated the intriguing possibility of obtaining β sheet formations in polymers which interact in two different ways. Depending on the modeling of the interactions we distinguish β sheets that align parallel or orthogonal to each other, which leads to two different phases. There remains an interesting theoretical question as to the behavior of the system when the parallel interactions are repulsive but the orthogonal interactions are highly attractive.

ACKNOWLEDGMENTS

Financial support from the Australian Research Council, the Centre of Excellence for Mathematics and Statistics of Complex Systems, and NSERC is gratefully acknowledged by the authors.

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